# Direct observation of melting and solidification of Bi1.8Pb0.4Sr1.9Ca2.1Cu3.5Ox in various oxygen atmospheres by high temperature optical microscopy 

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#### Abstract

The melting and solidification processes of the Bi 1.8 Pb 0.4 Sr 1.9 Ca 2.1 Cu 3.5 Ox in various oxygen atmospheres have been observed in situ using a high temperature optical microscopy. In the melting process, the samples began to melt at $788^{\circ} \mathrm{C}$ in $\mathrm{N} 2,862^{\circ} \mathrm{C}$ in $8 \% \mathrm{O} 2+\mathrm{N} 2,880^{\circ} \mathrm{C}$ in air, $895^{\circ} \mathrm{C}$ in O 2 . Many dark needle-like crystals $(\mathrm{Sr}, \mathrm{Ca})_{2} \mathrm{CuO} 3$ and the white bar-like crystals $(\mathrm{Sr}, \mathrm{Ca}) \mathrm{CuO} 2$ appeared in the liquid phase. When the temperature was raised, $(\mathrm{Sr}, \mathrm{Ca}) \mathrm{CuO} 2$ first melted and then $(\mathrm{Sr}, \mathrm{Ca})_{2} \mathrm{CuO} 3$ melted peritectically $((\mathrm{Sr}, \mathrm{Ca}) 2 \mathrm{CuO} 3 \rightarrow \mathrm{CaO}+\mathrm{L})$. In the solidification process, two peritectic reactions $\mathrm{CaO}+\mathrm{L} \rightarrow(\mathrm{Sr}, \mathrm{Ca})_{2} \mathrm{CuO} 3$ and $(\mathrm{Sr}, \mathrm{Ca})_{2} \mathrm{CuO} 3+\mathrm{L} \rightarrow \mathrm{Bi}-2212$ were observed in the atmospheres of air, $8 \%$ $\mathrm{O} 2+\mathrm{N} 2$, and O 2 , whereas in N 2 , only a reaction of $\mathrm{L} \rightarrow \mathrm{Bi}-2201$ was observed.


Key words: Bi-superconductor; Oxygen; Melting; Solidification; Microstructure

## 1. Introduction

[^0]Since Bi-Sr-Ca-Cu-O oxide was discovered to have a high transition temperature $\mathrm{Tc}[1]$, its preparation processes, structures, and superconducting properties have been investigated extensively. This system contains at least three kinds of superconducting phases, namely the $\mathrm{Bi}-2223$ ( Bi 2 Sr 2 Ca 2 Cu 3 Ox ) high-Tc phase, the $\mathrm{Bi}-2212$ ( $\mathrm{Bi} 2 \mathrm{Sr2CaCu2Ox}$ ) low-Tc phase, and the $\mathrm{Bi}-2201$ $(\mathrm{Bi} 2 \mathrm{Sr2CuOx})$ phase. The critical temperatures of these three phases are about $110 \mathrm{~K}, 80 \mathrm{~K}$ and 20 K , respectively. Although it has been found that partial substitution of Pb for Bi is favorable for obtaining the $\mathrm{Bi}-2223$ phase [2], the synthesis of single $\mathrm{Bi}-2223$ phase materials is still difficult due to poor understanding of its complex phase diagram, and solidification behavior [3,4]. Furthermore, the synthesis of Bi-2223 phase also critically depends on the atmosphere of the heat-treatments [5,6]. A reduced oxygen partial pressure was found to effectively enhance the efficiency of synthesizing Bi-2223 phase [5]. Recently [7], we reported the effect of oxygen partial pressure on the microstructure and formation of the $(\mathrm{Bi}, \mathrm{Pb})-2223$ phase in the partial-melting and sintering process. Samples with the composition Bi 1.8 Pb 0.4 Sr 1.9 Ca 2.1 Cu 3.5 Ox were prepared by following sintering conditions after partially melted at $875^{\circ} \mathrm{C}$ for 1 h in air: (i) $760^{\circ} \mathrm{C}$ for 120 h in N 2 , (ii) $825^{\circ} \mathrm{C}$ for 120 h in $8 \% \mathrm{O} 2+\mathrm{N} 2$, (iii) $840^{\circ} \mathrm{C}$ for 120 h in air and (iv) $845^{\circ} \mathrm{C}$ for 120 h in O 2 . In the samples sintered in pure N 2 and in pure O 2 , the $\mathrm{Bi}-2212$ phase as a major phase existed without the $\mathrm{Bi}-2223$ phase. The $\mathrm{Bi}-2223$ phase appeared in the samples sintered in air and in $8 \% \mathrm{O}_{2}+\mathrm{N} 2$, and the sample sintered in $8 \% \mathrm{O} 2+\mathrm{N} 2$ had the highest fraction of the Bi-2223 phase. In this study, the melting and solidification processes of the Bi 1.8 Pb 0.4 Sr 1.9 Ca 2.1 Cu 3.5 Ox in various oxygen atmospheres have been observed in situ using a high temperature optical microscopy.

## 2. Experiment details

High purity $\mathrm{Bi}_{2} \mathrm{O} 3, \mathrm{PbO}, \mathrm{SrCO}_{3}, \mathrm{CaCO}_{3}$ and CuO powders ( $99.99 \%$ in purity) were weighed and mixed in the atomic ratio $\mathrm{Bi}: \mathrm{Pb}: \mathrm{Sr}: \mathrm{Ca}: \mathrm{Cu}=1.8: 0.4: 1.9: 2.1: 3.5$, which is the optimum composition for the preparation of $(\mathrm{Bi}, \mathrm{Pb})-2223$ superconductor in the partial-melting and sintering processes in our previous experiment[8]. To increase sample's homogeneity, the initial powders were mixed and coarsely ground for 30 minutes in a ball mill with a little ethyl alcohol. The alcohol was then evaporated at $200^{\circ} \mathrm{C}$. The remaining sediment was calcined at $800^{\circ} \mathrm{C}$ for 12 h in a muffle furnace in air, and reground for 30 minutes in the ball mill. The calcined powders consisted mainly of the Bi-2212 phase. The film samples were prepared by coating the suspension of the calcined powders and ethanol onto a $\phi 5 \mathrm{~mm} \mathrm{MgO}$ disc. The melting and solidification of these film samples in various oxygen atmospheres $(\mathrm{N} 2,8 \% \mathrm{O} 2+\mathrm{N} 2$, air and O 2 ) at a heating and cooling rate of $2^{\circ} \mathrm{C} / \mathrm{min}$ were directly observed using a BHMJ-Olympus optical microscope with RHL-E-1R infrared rays condense furnace and Fujix HC-300Z digital camera.

## 3. Results and discussion

### 3.1. Melting process

Fig. 1 shows the optical microstructure changes of the calcined powders during heating in air. The micrograph in Fig.1a represents the starting materials. In the heating process, there was no apparent change below $870^{\circ} \mathrm{C}$ except for shrinkage by sintering. The sample began to melt at about $880^{\circ} \mathrm{C}$ and a lot of the dark needle-like phase and the white bar-like phase appeared in the liquid phase (Fig.1b). When the temperature was raised, crystals of the bar-like phase were firstly melted from $900^{\circ} \mathrm{C}$ to $920^{\circ} \mathrm{C}$, and those of the needle-like phase were melted from $960^{\circ} \mathrm{C}$ to $980^{\circ} \mathrm{C}$, but then the particle-like crystals appeared in the liquid phase (Fig.1c). Apparently, the sample was not melted perfectly at about $980^{\circ} \mathrm{C}$ (Fig.1d), where the particle-like phase existed in the liquid phase. According to analyses of SEM and EDS, the white bar-like crystals, dark needle-like crystals and the particle-like crystals are confirmed as $(\mathrm{Sr}, \mathrm{Ca}) \mathrm{CuO} 2,(\mathrm{Sr}, \mathrm{Ca}) 2 \mathrm{CuO} 3$, and CaO , respectively.

Figs. 2-4 show the optical microstructure changes of the calcined powders during heating in $\mathrm{N} 2,8 \%$ $\mathrm{O} 2+\mathrm{N} 2$ and O 2 , respectively. By comparison with the melting process in air, these in $\mathrm{N} 2,8 \% \mathrm{O} 2+\mathrm{N} 2$ and O 2 exhibit the same appearance but different melting temperatures. In the melting process, the samples began to melt at $788^{\circ} \mathrm{C}$ in $\mathrm{N} 2,862^{\circ} \mathrm{C}$ in $8 \% \mathrm{O} 2+\mathrm{N} 2$ and $895^{\circ} \mathrm{C}$ in O 2 , and a lot of the dark needle-like crystals $(\mathrm{Sr}, \mathrm{Ca})_{2} \mathrm{CuO} 3$ and the white bar-like crystals $(\mathrm{Sr}, \mathrm{Ca}) \mathrm{CuO} 2$ appeared in the liquid phase. When the temperature was raised, $(\mathrm{Sr}, \mathrm{Ca}) \mathrm{CuO} 2$ first melted and then $(\mathrm{Sr}, \mathrm{Ca})_{2} \mathrm{CuO} 3$ melted peritectically $\left((\mathrm{Sr}, \mathrm{Ca}){ }_{2} \mathrm{CuO}_{3} \rightarrow \mathrm{CaO}+\mathrm{L}\right)$.

### 3.2. Solidification process

Fig. 5 shows high-temperature optical micrographs of the sample during solidification in $8 \% \mathrm{O} 2+\mathrm{N} 2$. Above $950^{\circ} \mathrm{C}$, the liquid phase was in equilibrium with CaO (Fig.5a). From $940^{\circ} \mathrm{C}$ to $870^{\circ} \mathrm{C}$, the $(\mathrm{Sr}, \mathrm{Ca}) 2 \mathrm{CuO} 3$ formed in a needle-like shape by the peritectic reaction of the liquid phase and CaO , or by crystallizing directly from the liquid phase (Figs.5b and 5c). From $865^{\circ} \mathrm{C}$ to $860^{\circ} \mathrm{C}$, the plate-like 2212 phase was formed by the peritectic reaction of the liquid phase and $(\mathrm{Sr}, \mathrm{Ca})_{2} \mathrm{CuO} 3$. The solidification process finished at about $860^{\circ} \mathrm{C}$ (Fig.5d). Solidification process in air or in O 2 exhibited the same appearance but different solidification temperatures.

Fig. 6 shows high-temperature optical micrographs of the sample in the solidification process in N 2 . Above $780^{\circ} \mathrm{C}$, the liquid phase was in equilibrium with CaO (Fig. 6 a ). From $775^{\circ} \mathrm{C}$ to $770^{\circ} \mathrm{C}$, the plate-like 2201 phase was formed by crystallizing directly from the liquid phase (Figs.6b and 6c). Only a reaction of $\mathrm{L} \rightarrow \mathrm{Bi}-2201$ was observed during solidification in N 2 . The solidification process finished at about $770^{\circ} \mathrm{C}$ (Fig.6d).

## 4. Conclusions

The melting and solidification processes of the Bi 1.8 Pb 0.4 Sr 1.9 Ca 2.1 Cu 3.5 Ox in various oxygen atmospheres have been observed. In the melting process, the samples began to melt at $788^{\circ} \mathrm{C}$ in N 2 , $862^{\circ} \mathrm{C}$ in $8 \% \mathrm{O}_{2}+\mathrm{N} 2,880^{\circ} \mathrm{C}$ in air, $895^{\circ} \mathrm{C}$ in O . Many dark needle-like crystals $(\mathrm{Sr}, \mathrm{Ca}) 2 \mathrm{CuO} 3$ and the white bar-like crystals $(\mathrm{Sr}, \mathrm{Ca}) \mathrm{CuO} 2$ appeared in the liquid phase. When the temperature was raised, $(\mathrm{Sr}, \mathrm{Ca}) \mathrm{CuO}_{2}$ first melted and then $(\mathrm{Sr}, \mathrm{Ca})_{2} \mathrm{CuO} 3$ melted peritectically $\left((\mathrm{Sr}, \mathrm{Ca})_{2} \mathrm{CuO} 3 \rightarrow \mathrm{CaO}+\mathrm{L}\right) . \mathrm{In}$ the solidification process, two peritectic reactions $\mathrm{CaO}+\mathrm{L} \rightarrow(\mathrm{Sr}, \mathrm{Ca})_{2} \mathrm{CuO} 3$ and $(\mathrm{Sr}, \mathrm{Ca}) 2 \mathrm{CuO} 3+\mathrm{L} \rightarrow$ $\mathrm{Bi}-2212$ were observed in the atmospheres of air, $8 \% \mathrm{O} 2+\mathrm{N} 2$ and O 2 , whereas in N 2 , only a reaction of $\mathrm{L} \rightarrow$ Bi-2201 was observed.

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Fig. 1. High-temperature optical micrographs of the sample in the melting process in air:
(a) room temperature, (b) $890^{\circ} \mathrm{C}$, (c) $930^{\circ} \mathrm{C}$, (d) $980^{\circ} \mathrm{C}$

Fig. 2. High-temperature optical micrographs of the sample in the melting process in N 2 :
(a) $790^{\circ} \mathrm{C}$, (b) $800^{\circ} \mathrm{C}$, (c) $805^{\circ} \mathrm{C}$, (d) $807^{\circ} \mathrm{C}$

Fig. 3. High-temperature optical micrographs of the sample in the melting process
in $8 \% \mathrm{O} 2+\mathrm{N} 2$ : (a) $860^{\circ} \mathrm{C}$, (b) $875^{\circ} \mathrm{C}$, (c) $920^{\circ} \mathrm{C}$, (d) $960^{\circ} \mathrm{C}$
Fig. 4. High-temperature optical micrographs of the sample in the melting process in O 2 :
(a) $920^{\circ} \mathrm{C}$, (b) $950^{\circ} \mathrm{C}$, (c) $995^{\circ} \mathrm{C}$, (d) $1000^{\circ} \mathrm{C}$

Fig. 5. High-temperature optical micrographs of the sample in the solidification process in $8 \% \mathrm{O} 2+\mathrm{N} 2$ : (a) $955^{\circ} \mathrm{C}$, (b) $935^{\circ} \mathrm{C}$, (c) $925^{\circ} \mathrm{C}$, (d) $855^{\circ} \mathrm{C}$
Fig. 6. High-temperature optical micrographs of the sample in the solidification process in N 2 :
(a) $785^{\circ} \mathrm{C}$
(b) $774^{\circ} \mathrm{C}$, (c) $772^{\circ} \mathrm{C}$,
(d) $770^{\circ} \mathrm{C}$

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